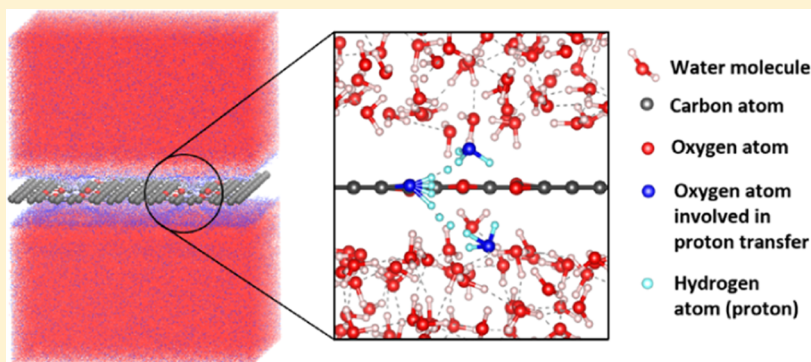


# Ether-Group-Mediated Aqueous Proton Selective Transfer across Graphene-Embedded 18-Crown-6 Ether Pores

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## Supporting Information



**ABSTRACT:** Nanoporous two-dimensional (2D) materials provide a new avenue for the design of zero-crossover proton selective membrane, which is critical for the development of direct methanol fuel cells and many other renewable energy systems. In this work, we investigate the aqueous proton selective conduction behavior across graphene-embedded 18-crown-6 ether pores using extensive ReaxFF molecular dynamics simulations. It is found that though there exists a vacuum gap between the aqueous phase and graphene membrane, the proton conduction behavior can be mediated by the crown ether functional groups and results in a low proton penetration energy barrier of  $5.53 \pm 0.29$  kcal mol<sup>-1</sup>, corresponding to a high proton conductivity of about  $3.23 \times 10^5$  S cm<sup>-2</sup>. Meanwhile, the vacuum gap together with the small pore size can effectively block the transportation of other molecules such as methanol, resulting in a high proton–methanol selectivity of about  $9.3 \times 10^{25}$ . Our results indicate that functional groups can significantly influence the proton conduction behavior across nanoporous 2D materials, and graphene membrane with embedded 18-crown-6 ether pores is a promising candidate for zero-crossover proton exchange membrane.

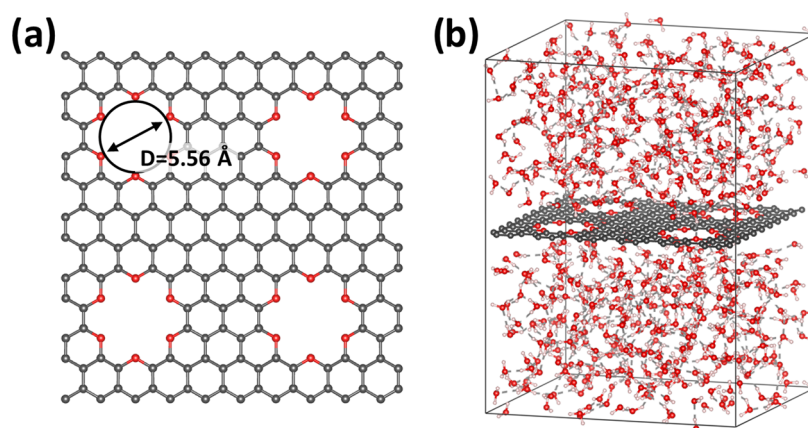
## 1. INTRODUCTION

Since the first discovery of graphene in 2004,<sup>1</sup> two-dimensional (2D) materials have become a new family of building blocks for membranes, due to their unique atomic structure.<sup>2–4</sup> By creating pores on 2D materials, selective mass transport at the nanoscale can be achieved, which has found broad applications in gas separation, water desalination, sensing, and so on.<sup>5–7</sup> In 2014, Geim et al. proposed that the uniform pores formed by the electron clouds of graphene and *h*-BN can allow the penetration of proton, while blocking the transportation of any other species.<sup>8</sup> This discovery opened up a new avenue for the design of zero-crossover proton exchange membrane and is particularly important for the development of direct methanol fuel cells (DMFCs), where methanol crossover can poison the noble metal catalyst in the cathode side, thus severely deteriorating the cell performance.<sup>9</sup> Follow-up experimental work shows that when sandwiching a single layer of graphene in Nafion membrane, the methanol crossover phenomenon can be greatly reduced.<sup>10–12</sup>

However, how protons conduct across 2D materials such as graphene and *h*-BN is controversial. Many theoretical<sup>13,14</sup> and experimental work<sup>15</sup> show that the energy barriers for aqueous proton conduction across pristine graphene or *h*-BN are too high to be realized at room temperature, and the previously observed proton conductivity may come from the contribution of atomic defects or bias potential. In our previous work, we investigated the relationship between the proton conduction capability and the pore size of nanoporous 2D graphyne.<sup>16</sup> We found that when the side length of the triangular pores in graphyne is smaller than 1.2 nm, a vacuum gap exists between the aqueous phase and graphyne, making proton conduction hard to be realized at room temperature. When the side length is larger than 1.2 nm, protons can transport via the Grotthuss mechanism along the “water wires” penetrating graphyne with high conductivity. It should be noted that graphyne is a

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**Figure 1.** Geometry of graphene-embedded 18-crown-6 ether pores in (a) vacuum and (b) an aqueous environment.

simplified model of nanoporous 2D materials with inert acetylenic pore edges, while in practical applications, the pore edges of nanoporous 2D materials can be decorated by different kinds of functional groups.<sup>17–19</sup> The relationship between functional groups and the proton conduction behavior is yet to be understood. When preparing nanoporous 2D materials, one of the most popular ways is to create nanopores on graphene.<sup>2</sup> Guo et al.<sup>20</sup> observed that the residual oxygen atoms in oxidized graphene form highly stable crown ether configurations within the 2D graphene layer. The graphene-embedded crown ether pores are predicted to show unique ion adsorption behavior towards  $K^+$  ions, making it a promising choice for ion-based logical elements<sup>21</sup> and strain-controllable ion sieving devices.<sup>7,22</sup> When serving as a proton exchange membrane, the presence of ether functional groups may change the hydrogen bond network in aqueous environment, thus influencing the proton conduction behavior. In this work, we explored the proton selective transfer behavior across graphene-embedded 18-crown-6 ether pores (as shown in Figure 1) using extensive reactive force field molecular dynamics (ReaxFF MD) simulations. The recently developed CHON-2017\_weak ReaxFF force field<sup>23,24</sup> was employed to accurately describe the weak interaction of functionalized hydrocarbon/water molecules in the condensed phase and capture the Grotthuss hopping proton motion.<sup>25</sup> From the ReaxFF MD simulations, we found that though the pore size is very small and there exists a vacuum gap between the aqueous phase and the graphene membrane, the ether functional groups can mediate the proton transfer process and significantly lower the proton conduction energy barrier to 5.53 kcal mol<sup>-1</sup>, which corresponds to a high area-normalized proton conductance of  $3.23 \times 10^5$  S cm<sup>-2</sup>. In the meantime, the vacuum zone together with the small pore size can effectively block the transfer of methanol molecules, providing an ultrahigh proton–methanol selectivity of  $9.3 \times 10^{25}$ . This proton selective conduction phenomenon cannot be observed in graphene pores terminated with carbon or hydrogen atoms. Our findings demonstrated the significant influence of functional groups toward aqueous proton transfer behavior and shed light on future designs of zero-crossover proton exchange membrane based on nanoporous 2D materials.

## 2. METHODOLOGY

**2.1. Density Functional Theory (DFT) Calculations.** All of the DFT calculations were performed using the Abinit

software package.<sup>26–28</sup> Perdew–Burke–Ernzerhof generalized gradient approximation<sup>29</sup> was adopted as the exchange–correlation functional, and the projector-augmented-wave method was used to describe the electron–ion interaction.<sup>30</sup> The cutoff energy was set to be 20 Ha and the  $k$ -point mesh was set to be  $<0.05 \text{ \AA}^{-1}$ . All of the structures were fully optimized to reach a force tolerance of  $0.01 \text{ eV \AA}^{-1}$ .

**2.2. ReaxFF MD Simulations.** The CHON-2017\_weak ReaxFF force field parameters<sup>23,24</sup> was used for all of the MD simulations. All MD simulations were performed using LAMMPS<sup>31,32</sup> software. For all production ReaxFF MD simulations, the time step is 0.25 fs, and the Nosé–Hoover chains thermostat was used with a pressure and temperature damping constant of 1000 and 100 fs for the NPT and NVT MD simulations, respectively. For metadynamics<sup>33</sup> simulations, PLUMED plugin<sup>34</sup> was used to deal with the proton hopping process. The Gaussian hills were deposited every 100 MD steps. The height and width of Gaussian hills were  $0.05 \text{ kcal mol}^{-1}$  and  $0.5 \text{ \AA}$ , respectively.

For proton penetration across the graphene membrane with embedded crown ether pores, proton hops between water molecules through the Grotthuss mechanism. The collective variable (CV) is defined as the distance  $L$  between the oxygen atom in the hydronium ion  $r_O(t)$  and the graphene membrane<sup>16,35,36</sup>

$$L = r_O(z) - g(z) \quad (1)$$

where  $r_O(z)$  can be calculated using

$$r_O(z) = \frac{\sum_{i \in \{O_w\}} z_i e^{\lambda n_i}}{\sum_{i \in \{O_w\}} e^{\lambda n_i}} \quad (2)$$

where  $z_i$  is the  $z$  position of waters' or hydronium's oxygen  $i$ ,  $\lambda$  is a large number (we set as 100 in our computation), and  $\{O_w\}$  refers to all oxygen atoms in the simulation system. The variable  $n_i$  is the hydrogen coordination number

$$n_i = \sum_{j \in \{H_w\}} n(r_{ij}) \quad (3)$$

where  $n(r_{ij}) = \frac{1 - (\frac{r_{ij}}{r_0})^6}{1 - (\frac{r_{ij}}{r_0})^{12}}$   $r_{ij}$  is the distance between oxygen atom

$i$  and hydrogen atom  $j$  and  $\{H_w\}$  refers to all hydrogen atoms in the simulation system. The variable  $r_0$  is set to be  $1.25 \text{ \AA}$ .

The variable  $n_i$  is very close to 3 when  $i$  is the oxygen atom of the hydronium ion and 2 in the case of a water molecule.

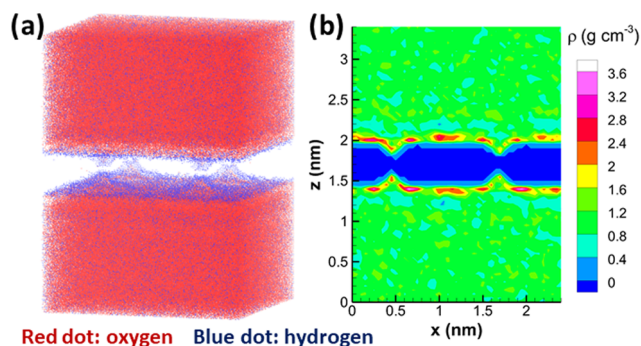
For methanol penetration, the CV was chosen as the distance  $L$  between the carbon atom in the methanol molecule  $r_C(t)$  and graphene

$$L = r_C(z) - g(z) \quad (4)$$

where  $r_C(z)$  and  $g(z)$  mean the position of the proton and the graphyne, respectively, in the  $z$  direction.

### 3. RESULTS AND DISCUSSION

**3.1. Graphene with Embedded Crown Ether Pores in Aqueous Environment.** The geometry of 18-crown-6 ether pores was taken from previous reports<sup>21</sup> and optimized using density functional theory (DFT) calculation. The model systems investigated in this work are based on a  $2.4 \times 2.5$  nm<sup>2</sup> graphene sheet containing four 18-crown-6 ether pores, as shown in Figure 1. The sheet was immersed in a periodic aqueous box with an initial height of 3.0 nm and water density of 1 g cm<sup>-3</sup>. Then, ReaxFF MD simulation in the NPT ensemble ( $P = 1$  atm,  $T = 300$  K) was performed for 2 ns with  $x$  and  $y$  directions fixed. Afterward, the system was equilibrated in the NVT ensemble ( $T = 300$  K) for another 2 ns. The detailed system configuration after equilibration can be found in Table S1. Then, we analyzed the water/graphene interphase by collecting data from another 1 ns ReaxFF MD simulation in the NVT ensemble. The water distribution surrounding the graphene membrane with embedded crown ether pores is shown in Figure 2a. It can be found that the aqueous phase and



**Figure 2.** (a) Water distribution surrounding the graphene membrane with embedded crown ether pores and (b)  $x$ - $z$  slice of the time-averaged three-dimensional water density at  $y$  corresponding to the center of the pore.

graphene membrane are separated by a vacuum zone. In the pore area, the vacuum zone narrowed but still exists. Figure 2b shows the  $x$ - $z$  slice of the time-averaged three-dimensional water density at  $y$  corresponding to the center of the pore. The height of the vacuum zone in the graphene area is about 5 Å and reduces to about 1.5 Å in the pore area. This water distribution is similar to the case of graphyne ( $n = 2$ ) in our previous work,<sup>16</sup> in which case protons cannot penetrate the membrane at room temperature, owing to the high energy barrier (16.37 kcal mol<sup>-1</sup>).

**3.2. Proton Selective Conduction Behavior.** The proton selective conduction behavior across the graphene membrane with embedded crown ether pores was studied by ReaxFF MD simulations. As proton selective conduction is particularly important for DMFC, methanol molecule was chosen to demonstrate the selectivity of graphene-embedded

crown ether pores. An extra proton was introduced into the aqueous environment, and 500 ps unbiased ReaxFF MD simulation in the NVT ensemble was conducted to equilibrate the system. The distance between the proton and the graphene membrane as a function of time during the equilibration process is shown in Figure S2. It can be found that no proton appeared in the vacuum zone surrounding the graphene membrane during the simulation time. Two water molecules in the aqueous system were replaced by a methanol molecule. Then 500 ps unbiased ReaxFF MD simulation in the NVT ensemble was conducted to equilibrate the system. The distance between the methanol molecule and the graphene membrane as a function of time during the equilibration process is shown in Figure S3. Similar to the case of the proton, no methanol molecule appeared in the vacuum zone surrounding graphene during the simulation time.

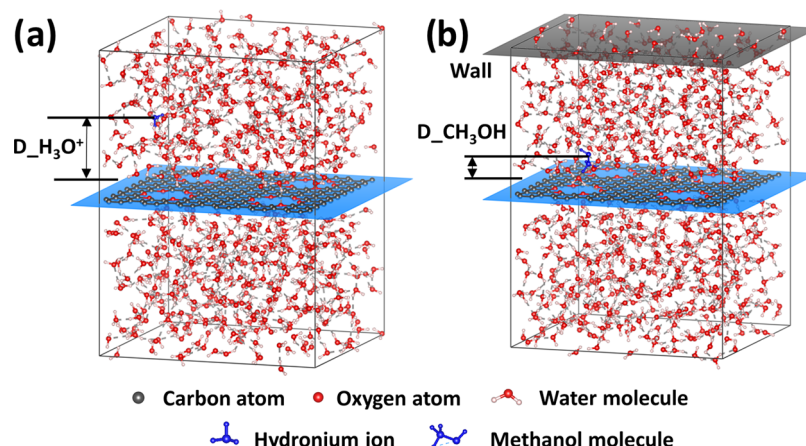
To obtain the proton and methanol penetration energy barriers, we conducted metadynamics simulations. The setup for metadynamics simulations is shown in Figure 3. For proton penetration, as protons mainly exist in the form of hydronium ion ( $H_3O^+$ ) in an aqueous environment, the distance between the oxygen atom in  $H_3O^+$  and graphene membrane was chosen to be the CV. The free energy profile was constructed once all phase space for the CV had been visited. For methanol penetration, the distance between the methanol molecule and the graphene membrane was chosen to be the CV. As the mobility of the methanol molecule is much lower than the proton, taking the same metadynamics setup with proton penetration will result in a large statistical discrepancy. To better evaluate the methanol penetration energy barrier, we took the metadynamics setup as shown in Figure 3b, where a wall 5 Å away from the top of the simulation box in the  $z$ -direction was added to restrain the methanol molecule movement. The free energy profile was constructed once the methanol molecule penetrated the crown ether pores.

For both proton and methanol penetration, we run three metadynamics simulations with different initial geometries. The energy barriers were calculated as the difference between the highest energy value and the lowest energy value. The detailed proton and methanol trajectory and energy profiles for each metadynamics simulation can be found in Figures S4–S15. The averaged energy barrier for proton conduction across the crown ether pores is  $5.53 \pm 0.29$  kcal mol<sup>-1</sup> as shown in Figure 4a, which is low enough to occur at room temperature. As the quantum effect of the proton is not considered here, the actual proton penetration energy barrier may be even lower.<sup>37</sup> For methanol penetration, the averaged energy barrier was calculated to be  $40.94 \pm 18.28$  kcal mol<sup>-1</sup>, which is about one order of magnitude higher than the case of proton and hard to be overcome at room temperature.

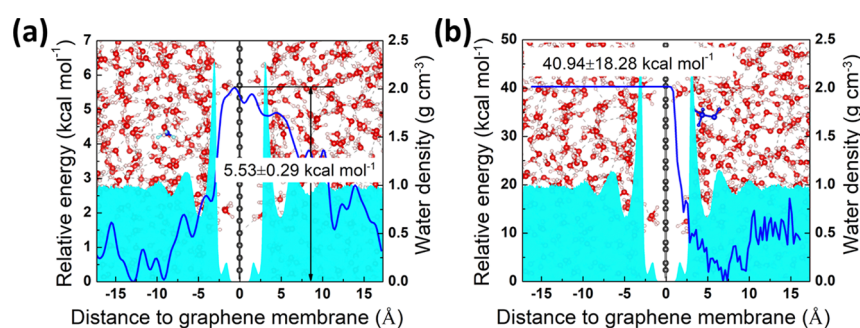
Based on the above calculated energy barriers, we can estimate the area-normalized proton conductance across graphene with embedded crown ether pores using the Nernst–Einstein<sup>38–43</sup> relation

$$\sigma_{H^+}^* = \frac{F^2}{RT} D_{H^+} C_{H^+} / d \quad (5)$$

where  $F$  is the Faraday constant,  $R$  is the ideal gas constant,  $T$  is the temperature,  $d$  is the membrane thickness, and  $C_{H^+}$  is the concentration of proton.  $D_{H^+}$  is the diffusion coefficient of the proton, which can be estimated using the Einstein–Smoluchowski equation<sup>40,42–44</sup>



**Figure 3.** Setup for metadynamics simulations of (a) proton and (b) methanol penetration across the crown ether pores.



**Figure 4.** Free energy profiles of (a) proton and (b) methanol as a function of distance to the graphene membrane with embedded crown ether pores. The shaded blue area represents water density and the background shows the initial geometries of the simulation system.

$$D_{H^+} = \frac{l^2}{\kappa\tau_D} \quad (6)$$

where  $l$  is the mean step distance and  $\kappa$  is a constant depending on the dimensionality of random-walk ( $\kappa = 2, 4,$  or  $6$  for one-, two-, and three-dimensional walk).  $\tau_D$  is the mean time between successive steps, which can be estimated as<sup>40</sup>

$$\tau_D = \nu_0^{-1} \exp\left(\frac{\Delta G}{k_B T}\right) \quad (7)$$

where  $k_B$  is the Boltzmann constant,  $\nu_0$  is the thermal frequency with  $\nu_0 = k_B T/h$ ,  $h$  is the Planck constant, and  $\Delta G$  is the effective Gibbs free energy of activation for proton diffusion, which we substituted with our calculated energy barriers.

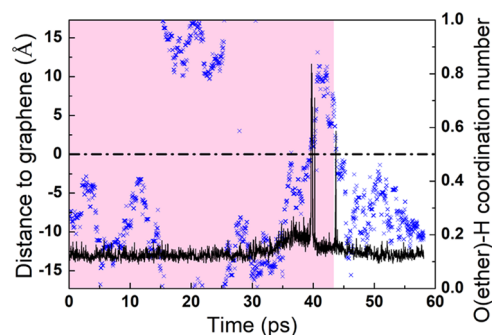
The height of the vacuum zone in the graphene area is  $5.0 \text{ \AA}$ , which is further used to estimate the membrane thickness and the mean step distance for proton penetration. Here, we take  $\kappa = 2$  for the penetration process; thus, for proton conduction across the graphene membrane with embedded crown ether pores at room temperature with proton concentration of  $1 \text{ M}$ , the estimated area-normalized proton conductance is  $3.23 \times 10^5 \text{ S cm}^{-2}$ . This conductivity is comparable with our previously predicted graphyne ( $n = 3$  and  $4$ ) and is much higher than Nafion.<sup>45</sup>

The selectivity of graphene embedded with crown ether pores is estimated using the Arrhenium equation<sup>2,46</sup> as

$$S \approx e^{-\frac{\Delta G_{H^+}}{k_B T}} / e^{-\frac{\Delta G_{CH_3OH}}{k_B T}} \quad (8)$$

Substituted with the values obtained from our ReaxFF MD simulations, the selectivity of the graphene membrane embedded with crown ether pores is  $9.30 \times 10^{25}$ . This value is high enough to achieve a zero-crossover proton exchange membrane for DMFC.

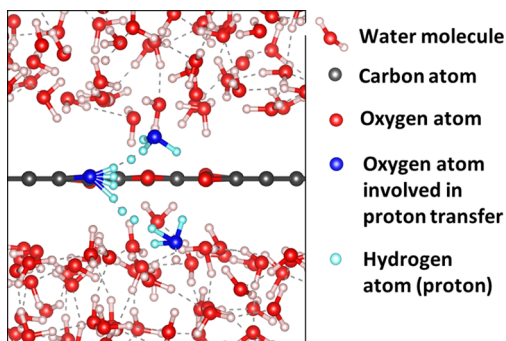
We now discuss the reason for this unexpected high proton conductivity. Figure 5 shows the distance between the proton and the graphene membrane together with the coordination number between the oxygen atoms in crown ether functional groups and the hydrogen element ( $O_{\text{ether}}\text{-H}$ ) as a function of metadynamics simulation time. It can be found that every time the proton penetrates the graphene membrane, the  $O_{\text{ether}}\text{-H}$



**Figure 5.** Distance between proton and the graphene membrane and coordination number between oxygen atoms in the crown ether functional groups and hydrogen element as a function of simulation time. The pink area represents the data used to construct free energy profile.

coordination number will increase dramatically from 0.1 to a high value between 0.6 and 0.9. This indicates that every time the proton penetrates the crown ether pores, it interacts with the ether functional group, which may help reduce the penetration energy barrier.

We analyzed the proton trajectory for the proton penetration process. It is found that every time the proton penetrates crown ether pores, it will approach the oxygen atom in the crown ether functional group, as illustrated in Figure 6.

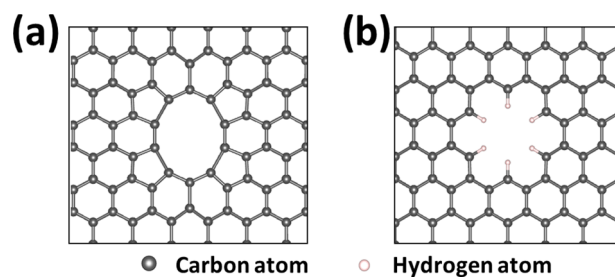


**Figure 6.** Proposed aqueous proton selective transfer mechanism across crown ether pores.

The crown ether functional group will serve as a mediator to collect proton from the hydronium ion on one side and pass it to a water molecule on the other side. No direct penetration of hydronium ions has been observed. This behavior significantly lowers the energy barrier for proton penetration across the graphene membrane compared with the cases where protons have to pass through the vacuum zone alone or drag a hydronium ion to the other side.

**3.3. Stability of Graphene Membrane with Embedded Crown Ether Pores.** In all of the above simulations, the position of the graphene membrane was fixed for the ease of data analysis. To demonstrate the stability of graphene with embedded crown ether pores in an aqueous environment, we performed 1 ns ReaxFF MD simulation in the NVT ensemble without any restraint. The snapshot after the simulation is shown in Figure S16. It can be found that the atomic structure of crown ether pores stays unchanged. Afterward, an extra proton was introduced into the system, and another 1 ns ReaxFF MD simulation without any restraint in the NVT ensemble was performed. The snapshot after the simulation is shown in Figure S17. The appearance of extra proton did not show any influence toward the stability of crown ether pores.

**3.4. Graphene Pores with Other Terminations.** To further demonstrate the function of crown ether functional groups, we investigate the proton transfer process across graphene pores with carbon or hydrogen atom as terminations, as shown in Figure 7. For graphene pores terminated with carbon atoms, six carbon atoms were removed from the graphene lattice similar to the case of crown ether pores. Detailed geometry was taken from the experimentally observed stable configuration<sup>47</sup> and optimized using DFT calculations. When putting it into the aqueous environment, water molecules around the pores will dissociate into hydroxyl groups and hydrogen atoms bonded with the carbon atoms in a short time within 5 ps, as shown in Figure S18, indicating that graphene pores terminated with carbon atoms are not stable in an aqueous environment.

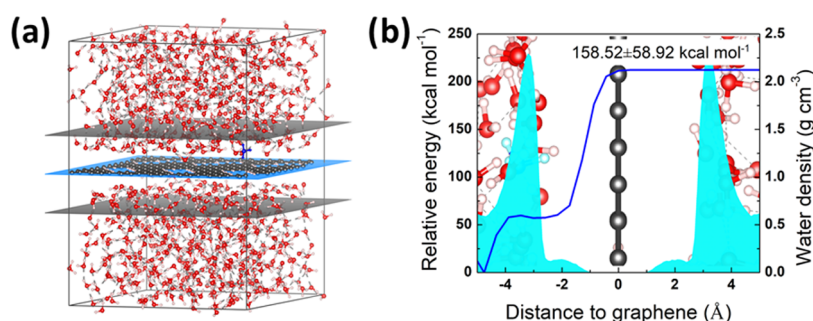


**Figure 7.** Geometries of graphene pores terminated with (a) carbon and (b) hydrogen atoms.

To construct graphene pores terminated with hydrogen atoms, the oxygen atoms in crown ether pores were replaced with carbon atoms, and each carbon atom was bonded with a hydrogen atom. Then, we studied the proton penetration behavior across the graphene pores using metadynamics simulations. As the proton penetration energy barrier across the graphene pores terminated with hydrogen atoms are very high, when using the same metadynamics simulation setup as the crown ether pores, no proton penetration behavior can be captured when the deposited bias potential reaches as high as 25 kcal mol<sup>-1</sup>. Due to the complexity of the CV used, further deposition of bias potential will result in the self-dissociation of water molecules and no meaningful results can be obtained. To better evaluate the penetration energy barrier, we took the metadynamics simulation setup similar to our previous work when dealing with graphyne ( $n = 1$ ), as shown in Figure 8a.<sup>16</sup> All of the O–H bonds in water molecules were constrained when the bond length exceeded 1.4 Å, and a pair of walls were put 5 Å away from the graphene membrane to confine the proton movement. The distance between the extra proton and graphene was used as CV. The energy profiles were constructed once the proton penetrates the graphene pores from one to the other side. The detailed proton trajectory and energy profiles for each metadynamics simulation can be found in Figures S19–S24. The averaged energy barrier for proton conduction across the graphene pores terminated with hydrogen atoms is 158.52 ± 58.92 kcal mol<sup>-1</sup>, as shown in Figure 8b, which is about 30 times of that for proton penetration across crown ether pores.

## 4. CONCLUSIONS

In this work, we investigated the proton selective conduction behavior across graphene-embedded 18-crown-6 ether pores using extensive ReaxFF MD simulations. It is found that though there exists a vacuum gap between the aqueous phase and graphene, the proton conduction behavior can be mediated by the crown ether functional groups, leading to a low proton penetration energy barrier of 5.53 ± 0.29 kcal mol<sup>-1</sup>. In the meantime, the vacuum gap as well as the small pore size can effectively block the penetration of other molecules such as methanol. The proton conductivity for graphene with embedded crown ether pores is estimated to be 3.23 × 10<sup>5</sup> S cm<sup>-2</sup>, and its selectivity towards methanol is estimated to be 9.30 × 10<sup>25</sup>. Our work demonstrated the significant influence of pore edge functional groups on the proton selective conduction behavior and proposed that graphene with embedded 18-crown-6 pores can provide a high proton conductivity and an ultrahigh selectivity simultaneously, which shed light on the future design of zero-crossover proton exchange membrane.



**Figure 8.** (a) Setup for metadynamics simulations of proton penetration across hydrogen-terminated pores and (b) free energy profiles of proton as a function of distance to graphene membrane. The shaded blue area represents water density and the background shows the initial geometries of the simulation system.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpcc.9b09715.

Supplementary details of the simulated systems; methods and additional results and discussion (PDF)

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### Notes

The authors declare no competing financial interest.

## ■ ACKNOWLEDGMENTS

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